ABSTRACT

Continued tightening of regulations on hexavalent chromium used in decorative plating have caused an increased interest from original equipment manufacturers (OEMs) in the performance of trivalent chromium plating with regards to extended outdoor corrosion as well as accelerated corrosion performance tests. This presentation reports on corrosion testing results of trivalent chromium deposits with various thickness levels of chromium and various STEP values of the underlying nickel deposits.

Introduction

Among the various original equipment manufacturers (OEMs), there has been an overall curiosity about how trivalent chromium performs versus hexavalent chromium in corrosion tests with different chromium thicknesses and with different STEP† values in the underlying nickel. This paper reviews those results. All test results are from a trivalent chromium-mixed sulfate/chloride electrolyte versus a dual catalyst hexavalent chromium (fluoride/sulfate) electrolyte.

Basis for the test matrix

Trivalent chromium is microporous at low thicknesses, and micro-discontinuous (i.e., microporous or microcracked) at high thicknesses. In contrast to hexavalent chromium deposition, trivalent chromium needs no particles in the underlying nickel layer to achieve porosity. However, too high a thickness can lead to microcracking. The OEM request was to identify the thickness effect on performance, given the structural difference over the thickness range.

With trivalent chromium, corrosion resistance is enhanced by a STEP potential between the bright nickel and the low sulfur strike. The proper STEP value prevents undercutting of the trivalent chromium deposit. In this regard, the OEM request related to identifying the effect and the range for best performance.

OEM test matrix - Phase 1

The objectives of the Phase 1 test matrix were threefold:

1. Evaluate trivalent chromium versus hexavalent chromium at 40 mV.

† Simultaneous Thickness and Electrochemical Potential (STEP) test, measuring the potential between layers of a multilayer nickel plating system.
2. Evaluate trivalent chromium at three different thickness values, 0.15, 0.3 and 0.5 microns over microporous nickel (extremely high porosity).

3. Evaluate hexavalent chromium over a low sulfur nickel strike (LSS) with extremely low porosity, for reference.

The test matrix variables, as noted above, were the chromium thickness and STEP potential measurements. The change in character of trivalent chromium from microporous at low thickness to micro-discontinuous at higher thickness suggests that the corrosion performance would be affected by the transition. Hence, there was a need to identify the thickness effect on performance.

**STEP testing**

The STEP Test, developed by Harbulak in 1980,¹,² measures the potential difference between the nickel plating layers, in a decorative multilayer system, e.g., between semi-bright and bright nickel layers in a dual nickel system. Measured and reported in millivolts (mV), the values can be either positive or negative. The importance lies in whether the nickel layer is cathodic or anodic with respect to the reference layer. More active layers will preferentially corrode as compared to more noble layers.

Figure 1 relates the STEP test to a modern four-layer nickel system. The photo compliments the schematic at the left. Starting at the copper substrate and working up, there is a semibright nickel layer, a high sulfur layer, a bright nickel layer, a microporous layer (or low sulfur strike) and finally a thin chromium layer (too thin to be visible). The photo shows a small corrosion area on the top with the corrosion proceeding through the bright nickel layer and then spreading out horizontally at the high sulfur layer. The system is designed to control the corrosion to provide a visually defect-free coating after extended corrosion.

The graph at the right shows an ideal STEP curve for the four-layer nickel system. The zero point on the x-axis corresponds to the copper-nickel interface. Working upward the bottom dashed arrow points to (1) the semi-bright Ni potential (line H-I), (2) the thin high sulfur nickel potential (point E), (3) the bright nickel potential (line CD) and (4) the microporous nickel (point B). In these tests, the STEP values for Phase 1 testing were maintained at:

- Semibright to bright  110-130 mV difference
- High sulfur to bright    20 - 40 mV difference
- Microporous to bright    30 - 50 mV difference

---


Corrosion testing

Evaluation of corrosion performance is based on (1) the number of active sites (i.e., porosity) observed after 88 hr of CASS or Porosity testing and (2) 336 hr of the ASTM-B-995 Russian Mud Test. Figure 2 shows the types of active sites encountered in the Phase 1 corrosion testing after 88 hr of CASS exposure. Low porosity (at left) will concentrate the corrosion current in a few areas, causing deep wide corrosion pits visible to the eye after extended outdoor exposure. “Normal” porosity, between the two extremes shown, will spread out corrosion to many small areas, making the corrosion invisible to the eye. High porosity (at right) can make corrosion areas so frequent and small that it can become visible to the eye as a haze.

Phase 1 test results

The ratings of the Phase 1 test panels after 88 hr of CASS testing are shown in Fig. 3. From left to right, the first column shows the nickel type directly under the chromium, whether it is a microporous nickel to provide the STEP potential and porosity from particles in the nickel or a low sulfur strike to simply provide an appropriate STEP value. The next column shows the STEP value in mV, followed by the thickness of the chromium in μm. Next is the type of chromium whether hexavalent (HEX) or trivalent (TVC). The last two columns show the panel ratings, 1-10, for visible haze and surface pits.

Figure 4 shows the ratings of the panels after 336 hours of Russian mud testing. The configuration of the table is similar to that for the CASS testing.

Phase 1 discussion

The corrosion protection provided by the decorative chromium overlay operates somewhat differently, when the hexavalent and trivalent chromium are compared. In the hexavalent case (Fig. 5(a), the chromium provides hard passive layer and white color. The microporous nickel provides porosity and prevents undercutting of the chromium layer. The sulfur-containing bright nickel acts as the sacrificial layer. The high-sulfur nickel is anodic to the bright and semi-bright layer, while the semi-bright nickel layer prevents corrosion to the substrate.

Considering the trivalent chromium case (Fig. 5 (b), every layer from the copper up to the bright nickel is the same as the
hexavalent chromium system. But the layer above the bright nickel is commonly called “low sulfur nickel.” Its name is different because there is no need for particles to induce porosity when using trivalent chromium. The trivalent chromium provides hard passive layer and white color, as does hexavalent chromium, but also is microporous as plated.

The obvious operational advantage of using trivalent chromium is the fact that the porosity is uniform over the entire surface. There is no need to shut down the line to treat the nickel (remove the particles) and no fear of reduced porosity from high chromium thickness.

A comparison of the active sites produced after 88-hr CASS for the systems shown in Fig. 5 is shown in Fig. 6. Here, 0.3 μm of trivalent chromium over low sulfur nickel is compared with 0.3 μm of hexavalent chromium over microporous nickel, corresponding to the data of samples E and G referenced in Fig. 3, respectively.

**Figure 5** - Comparison of (a) hexavalent and (b) trivalent chromium overlays on four-layer nickel.

**Figure 6** - Comparison of active sites after 88-hr CASS for (a) trivalent and (b) hexavalent chromium over nickel.

The trivalent result (L) shows a relatively uniform dispersal of active corrosion sites, with a narrow range of pore diameters, suggesting the corrosion current is well distributed. On the other hand, fewer sites are seen for the hexavalent result (R), with a wider range of pore diameters. The corrosion current distribution is less uniform and, in some cases, the active sites were large enough to produce heavier local nickel attack. Nonetheless, the CASS corrosion ratings for both samples (Fig. 3, E and G) were good. Analysis of the 336-hr Russian mud data for chloride resistance from Fig. 4 for conditions E and G shows that the trivalent system (Rating = 3) performed better than the hexavalent (Rating = 1).

Referring again to the 88-hr CASS data in Fig. 3, three trivalent chromium samples with increasing thicknesses, 0.15, 0.30 and 0.50 μm (H, J, and L, resp.) are shown. In this case however, the top nickel layer was the microporous nickel usually used with hexavalent chromium. This creates very high porosity in the chromium layer. The low haze ratings in the data reflect this result. The corresponding Russian mud rating did not change with thickness (Fig. 4, H, J and L).

Conversely, plating hexavalent chromium over a low sulfur nickel deposit tends to create very low porosity. The surface pit rating (Fig. 3, K) reflects this. The 88-hr CASS result in Fig. 8 is a striking example of the localized corrosion current attack developed in this case.
Future work

Additional work planned with the OEM Test Matrix is shown in Fig. 9 with emphasis on 48-hr CASS results, as well as additional Russian mud testing. The trivalent chromium-low sulfur nickel work will focus on varying the step potential and trivalent chromium thickness.

Work on the horizon will include a similar test matrix with black trivalent chromium, and test matrices involving functional deposits, including (1) zinc-nickel with a black passivate and sealer and (b) zinc-nickel for ductility.

About the author

Mark Schario is Executive Vice-President at Columbia Chemical Corporation, headquartered in Brunswick, Ohio. He joined Columbia Chemical in 2010 as Vice President, Technologies and now serves as Vice President, Global Business Development for the company. He has over 30 years of experience in the surface finishing industry. Mark also functions as the company’s top liaison to the automotive industry and is Committee Chairman to ASTM B08 on Metallic and Inorganic Coatings which has jurisdiction over 132 standards. He is a member of the NASF and has earned the industry designation of CEF/Certified Electroplater-Finisher. Mark holds an Executive MBA from the Weatherhead School of Management at Case Western Reserve University.